Dry Sliding Wear Behavior of B₄C Particulates Reinforced Al7020 AlloyComposites

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Abstract

The work is carried out to investigate the dry sliding wearbehaviorofB₄CreinforcedAl7020alloymetalmatrixcompos ites. In the present work Al7020 alloy was taken as thebase matrix and B₄C particulates as reinforcement material toprepare metal matrix composites by stir casting method. Formetalmatrixcompositesthereinforcementmaterialwasvaried 2wt. from 0to4wt. % steps in of %. Thewearresistanceofmetalmatrixcompositeswasstudiedbyperf orming dry sliding wear test using a pin on disc apparatus. The experiments were conducted at a constant sliding speed of300rpm and sliding distance of 4000m over a varying load of1, 2 and 3Kg. Similarly, experiments were conducted at aconstant load of 3Kg and sliding distance of 4000m over avarying sliding speed of 200, 300 and 400rpm. The resultsshowed that the wear resistance of Al7020-2% B₄C and 4%B₄C composites were better than the unreinforced alloy. Thewear in terms of height loss found to increase with the loadandslidingspeed.Tostudythedominantslidingwearmechani sm for various test conditions, the worn surfaces wereanalyzedusingscanningelectron microscopy.

Keywords: Al7020 Alloy, B₄C Particulates, Microstructure, Wear, WornSurface

INTRODUCTION

Metalmatrixcomposites(MMCs)havebeenacceptingextraordin aryconsiderationbecauseoftheirhighrigidity,hardness

andmodulus, and also their high wear resistance contrasted with framework. tribological properties the The ofaluminumcompoundscanbealtogetherupgradedbytheaddition ofadispersedceramicparticlephase[1,2].Thesubsequent material is all around known as aluminum matrixcomposites (AMCs). The improved properties of AMCs, forexample, high particular quality, lessened wear rate and lower warm extension has pulled in the consideration of the materialsdesigninggroup.AMCsarereplacingregularaluminum alloys in applications aviation. few including а car, shipbuilding and atomic designing [3]. Among different ceram ic particles B₄C is progressively favored as support for AMC sbecause of its thermodynamic dependability, high dissol point. high hardness, high flexiblemodulus ving andremarkablewear resistance.

AMCs could be delivered either by solid state or fluid (liquid)state preparing [4, 5]. Liquid state systems are for the

mostpart utilized in light of the fact that these are economicallyviable,simpleandapplicableinlargequantityprodu ction.Melt stirring techniques are of two sorts in particular ex situandin-situ.In'ex-

situ'procedureartisticfortificationsareadded remotely to the liquid metal though 'in-situ' handle thegeneration of support happens inside the network accordinglyof compound response. The in-situ composites represent a fewfocal points over ex-situ, for example, uniform dispersion offortificationparticles,grainrefinement,clearinterface,improve dwarmsteadinessand prudentpreparing[6-8].

Al-ZnbasedAl7020allovisheattreatablealuminumcomposite, which offers high quality at low particular weightand are broadly utilized as basic segments, especially in theaerospace industry. B₄C strengthened Al7020 composites aretherefore be consideredto havemore application potential. ItishoweveratestingundertakingtocreatehomogeneousAl/B₄CA MCsutilizingthesestrategies.Regularimperfections, for example, porosity, agglomeration, isolationand slag consideration are for the most part unavoidable. Ahomogeneous dispersion of B₄C particles is fundamental toaccomplishupgraded properties and execution.

Over the last decade, a lot of studies have been carried out toovercometheseinsufficienciesandplentyofexperimentalprepa rations have been performed. Several efforts have beenmade to improve the interface bonding between aluminiumandceramicparticlesbyexploringsurfacecoatingofthe ceramicparticles.InthepresentstudyAl7020-

 $B_4C composites have been fabricated by using novel two stepmixing process of reinforcements, to improve the wettability of B_4C particulates with Alalloy matrix.$

In this study, an attempt has been made to prepare Al7020alloy composites by adding 2 & 4 wt. % of B₄C particulatesinto matrix by using a novel two stage reinforcement additionmethod. Further, the prepared Al7020 – B₄C composites werestudied for effect of load and sliding speed on the wear propertie sby using pin-on-disc weartesting machine.

EXPERIMENTALDETAILS

Metal matrix composites containing 2 and 4 weight rates $ofB_4Cparticles were created by liquid metallurgy course. For$

the generation of MMCs, an Al7020 alloy was utilized as the frameworkmaterialwhileB₄Cwereutilizedasthefortifications . The theoretical density of grid material Al7020 amalgamis2.80g/cm³ and support particulates B₄Cis2.52g /cm³. The chemical substance of Al7020 composite utilized as a part of the work is given in the table 1.

Table1. Chemical composition of Al7020 alloy

Elements (symbol)	Zn	Cu	Mn	Mg	Fe	Cr	Si	Ti	Al
Wt.per centage	5	0.2	0.5	1.2	0.4	0.3	0.35	0.1	Bal

TheB₄CparticlereinforcedAl7020alloymetalmatrixcomposites have been produced by using a vortex method. Initially calculated amount of Al7020 alloy was charged intoSiC crucible and superheated to a temperature 730°C in anelectrical resistance furnace. The furnace temperature was contr olled to an accuracy of ±10 degree Celsius using a digital temperature controller. Once the required temperature is a characteristic temperature ieved, degassing is carried out using solid hexachloroethane (C_2Cl_6) to expel all the absorbed gases. Themeltwasagitatedwiththehelpofazirconiacoatedmechanical stirrer to form a fine vortex. A spindle speed of 300 rpm and stirring time 3-5 min. were adopted. The B₄Cparticulates were preheated to a temperature of 500 degreeCelsius in a preheater to increase the wettability. The pre-heated B₄C particles introduced into melt in steps of two atconstant feed rate of 1.2-1.4 g/sec. After holding the melt for aperiod of 5 min., the melt was poured from 710 degree Celsiusinto a preheated cast mould having dimensions of 120mmlength iron x15mmdiameter.

Metallographictestspecimensof5mmthicknesswereprepared by cutting the as cast and B_4C strengthened Al7020combinationcomposites.Testsampleswerepolishedacco rdingtothestandardmetallographicmethodology and etched with Keller's reagent. The microstructure was viewedutilizingscanningelectronmicroscopeinstrument.

The dry sliding wear behavior of as cast A17020 alloy andA17020-B₄C composites were evaluated using a pin-ondiscwear apparatus at room temperature according to ASTM G99standard.Pinsoflength25mmanddiameter8mmwereprepare dfromthecastsamples.Theexperimentswereconducted at a constant sliding speed of 300rpm and slidingdistance of 4000m over a varying load of 1Kg, 2Kg, and 3Kg.Similarly experiments were conducted at a constant load of3Kg and sliding distance of 4000m over a varying slidingspeed of 200, 300 and 400rpm. The polished surface of the pinwas slide on a hardened chromium steel disc. A computeraided data acquisition system was used to monitor the loss ofheight.Wearvalueispresented intermsofheightloss.

RESULTSANDDISCUSSION

Microstructurestudy







Fig. 1. a-c Showing the scanning electron microphotographs of (a) as cast Al7020 alloy (b) with 2 wt.% of B_4C & (c) with4wt.% of B_4C

Figure 1 (a-c) shows the SEM microphotographs of Al7020alloyascastandAl7020with2and4wt.%ofB₄Cparticulate composites. Thisreveals theuniform distribution B₄C particles and very low agglomeration and segregationofparticles, and porosity.

Fig. 1 b-c clearly show and even distribution of B_4C particles in the Al7020 alloy matrix. In other words, no clustering of B_4C particle is evident. There is no evidence of casting defects uch as porosity, shrinkages, slag inclusion and cracks which is indicative of sound castings. In this, wetting effect

betweenparticles and molten Al7020 alloy matrix also retards the m ovement of the B_4C particles, thus, the particles can remain suspended for a long time in the melt leading to uniform distribution.





Fig.2: EDSspectrumof(a) ascastAl7020 alloy (b) Al7020-4%B₄Ccomposites

In order to confirm the presence of B_4C energy dispersivespectroscope analysis was carried out at the edge of the B_4C particle and Al alloy matrix. The EDS spectrum reveals the presence of Al, Zn, Cu, Mg, B and C in the interface reactionlayer (fig. 2-b).

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EffectLoadonWear

The variation of wear loss at steady 300rpm sliding speed andchanging loads of 1Kg, 2Kg and 3Kg is as appeared in fig. 3.Applied load influences thewear ofAl7020 compound andthe composites fundamentally and is the most overwhelmingcomponent controlling the wear conduct. The wear misfortunechanges with the typical load and is altogether lower if thereshould arise an occurrence of composites. With increment inloads there is higher wear misfortune for matrix alloy and

thecomposites.Howeveratalltheloadsconsideringwearresistanc eofthecompositesisbetterthantheframeworkcombination. At higher loads and the transition to sever wearthe surface temperature exceeds a critical value. So as appliedload increases ultimately there is an increase in the wear lossfor both the reinforced and unreinforced composite materials.Thevariationofwearlossofthematrixalloyanditscomp osites with 2 and 4 wt. % of B₄C content is shown in fig.3.

The improvement in the wear resistance of the composites with B_4C reinforcement can be attributed to the improvement in the hardness of the composites and improved hardn essresults in the decrease in the wear loss of the composites [9,10].



Fig. 3: Showing wear of Al7020 alloy and its composites atvaryingloadsand sliding speed of300rpmand 4000mslidingdistance

EffectSlidingSpeedonWear

Fig. 4 shows the variation of wear loss of A17020 matrix alloyand Al7020-2% & 4% B₄C composites at constant 3Kg loadand varying sliding speeds. With an increasing speed i.e. 200,300, and 400 rpm, there is an increase in thewear loss forbothmatrixalloyanditscomposites.Howeveratallthesliding speeds studied, the wear loss of the composite wasmuch lower when compared with the matrix alloy. Furtherincreasedwear ratewith increasedsliding speedis duetothermal softening of composite. On the the other hand theincreasedtemperatureathigherslidingspeedscancausesevere plastic deformation of the mating surfaces leading toformhighstrainratesub-surfacedeformation[11,12].The

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increasedrateofsub-

surfacedeformationincreasesthecontactareabyfracture,andfrag mentationofasperities.Therefore this leads to enhanced delamination contributing toenhancewearrate.Further,4wt.%ofB₄Cparticulatesreinforced Al7020 alloy composites shown more resistance towear.



Fig.4:ShowingwearofAl7020alloyanditscompositesatvaryi ng sliding speed, constant load of 3Kg and 4000mslidingdistance

WornSurfaceMorphology





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(c)

Fig.5:ShowstheSEM microphotographsofwornsurfaces of (a) ascastAl7020 alloy(b)Al7020– 2wt.%B₄C (c)Al7020–4wt.%B₄Ccompositesat3Kgload and300rpm

Wearsurfaceanalysis of composites were examined by scanning electron microscope. Fig.5a-c represents the wearsurface of as cast A17020 alloy and specimens containing 2 &4 wt. % of B_4C particles reinforced composite at 3Kg load and 300 rpmsliding speed.

The examination of worn surface as appeared in figure 4a, thatthewornsurfaces of base combination are considerably rougher thancomposites.Cavitiesandextensivefurrowedsurfaces are found on worn surface of Al7020 amalgam. Thesign of pits and scores underpins the way that delicate Alcomposite load disfigured at higher of 3kg and at 300rpmspeedandhauledoutfromthesurface. The wear trackperce ptiondemonstratesthatgripanddelaminationareprevailingwearin strumentsseenathigherloads. This is supported by the large sized delamination flakes and severeadhesion resulting in bulk removal of material at higher loads[13].

Fig. 5b & c shows the SEM image of the worn surface of Al7020-2 & 4wt.% of B_4C composite tested at applied load of 3kg and 300rpm speed. The grooves are very small due to the hard nature of B_4C reinforcement and poor wear losses. As the ceramic particles resist the delamination process, composite s are found to have greater wear resistance [14]. Worn surface shows less cracks and grooves mainly due to the presence of hard particulates.

CONCLUSIONS

The present work on processing and evaluation of Al7020-B₄Cmetalmatrixcompositebymeltstirringhasledtofollowing conclusions. Al7020 alloy based composites havebeen successfully fabricatedby meltstirringmethodusingtwo stage combined method reinforcement addition of withpreheating of particles. The SEM microphotographs of compositesrevealedfairlyuniformdistributionofreinforcementparticula tesintheA17020metalmatrixandEDS spectrographs confirmed the presence of B_4C $particles. The addition of B_4 C particles to Al alloy matrix improves t$ he

wear resistance of the composite. The wear loss is dominatedby load factor and sliding speed. The increase of loads andsliding speeds leads to a significant increase in the wear loss. The Al7020-4% B₄C composites have shown lower wear lossascompared to that observed inascastAl7020 alloyand2 wt.

%B₄Creinforcedcompositesmatrix.Wornmorphology_{showed} the effect of hard ceramic particulates addition onwearbehavior of Alalloyand itscomposites.

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